Nonequilibrium processes from Generalised Langevin Equations: realistic nanoscale systems connected to two thermal baths

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We extend the Generalised Langevin Equation (GLE) method [Phys. Rev. B 89, 134303 (2014)] to model a central classical region connected to two realistic thermal baths at two different temperatures. In such nonequilibrium conditions a heat flow is established, via the central system, in between the two baths. The GLE-2B (GLE two-baths) scheme permits us to have a realistic description of both the dissipative central system and its surrounding baths. Following the original GLE approach, the extended Langevin dynamics scheme is modified to take into account two sets of auxiliary degrees of freedom corresponding to the mapping of the vibrational properties of each bath. These auxiliary variables are then used to solve the non-Markovian dissipative dynamics of the central region. The resulting algorithm is used to study a model of a short Al nanowire connected to two baths. The results of the simulations using the GLE-2B approach are compared to the results of other simulations that were carried out using standard thermostatting approaches (based on Markovian Langevin and Nose-Hoover thermostats). We concentrate on the steady state regime and study the establishment of a local temperature profile within the system. The conditions for obtaining a flat profile or a temperature gradient are examined in detail, in agreement with earlier studies. The results show that the GLE-2B approach is able to treat, within a single scheme, two widely different thermal transport regimes, i.e. ballistic systems, with no temperature gradient, and diffusive systems with a temperature gradient.

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I. INTRODUCTION

Nanoscale devices and materials are becoming increasingly important in the development of novel technologies. In most applications of these new nanotechnologies, the central system is part of a more complex set up where driving forces are present to establish heat and/or particle flows. The understanding of the corresponding non-equilibrium properties is of utmost importance. This is especially true when one considers potential applications based on the thermal conductivity of materials¹⁰¹⁵ and the heat transport within nanodevices⁹¹¹⁵.

Being able to describe the dynamics and dissipation of such nanoscale atomic systems is central for modern nanoscience. For that, one has to consider the central region of interest as an open system surrounded by a heat bath (an environment) which is in contact with the system and is kept at a given temperature. For studying the transport properties, one has to consider the proper non-equilibrium conditions, i.e. the central region is connected to two (or more) independent heat baths (kept at their own temperatures). Hence a heat flow (transient or stationary depending on the experimental conditions) is established between the two baths via the central region.

An appropriate general approach for treating this kind of systems is based on the so-called Generalised Langevin Equation (GLE)¹¹¹²¹⁵. The GLE is an equation of motion containing non-Markovian stochastic processes where the particle (point particle with mass) has a memory effect to its velocity.

The GLE has been derived for a realistic system of N particles coupled to a single realistic (harmonic) bath, i.e. a bath described at the atomic level¹². The non-Markovian dynamics is obtained for the central system with Gaussian distributed random forces and a memory kernel that is exactly proportional to the random force autocorrelation function¹². Solving the GLE for complex heterogeneous and extended systems is still a challenge. A major step towards the solution of this problem for a realistic application has been recently given.¹⁷¹¹¹⁵ In particular, an efficient and transferable algorithm has been developed in Ref. ⁵² to solve the GLE numerically while taking into account the two fundamental features of the GLE – a time-dependent memory kernel and the presence of a coloured noise, which are absolutely essential for the description of the bath at the atomic level.

However, the previous tools have been developed for a single bath only. In order to treat properly the presence of a (transient or steady) flow of heat current through the central system, one has to take into account the proper non-equilibrium conditions. That is, one has to consider the presence of at least two baths (at their own temperatures) in contact with the central region.

The aim of the present paper is to extend the previous GLE approaches developed in Refs. ⁵² and ⁵³ to the systems consisting of a central region connected to two spatially separated thermal baths. With such an approach, we can study the non-equilibrium processes in nanoscale systems by using molecular dynamics (MD) simulations. The dissipative processes are
properly described since the system can exchange energy (heat) with the environment. The environment consists of the two baths whose dynamical properties are described more thoroughly than when standard thermostats (i.e., Langevin, Nose-Hoover, velocity renormalisation, etc.41,59) are used in conventional MD simulations. Note that, for a system at equilibrium, the Langevin dynamics applied only to a part of the system can be derived from the GLE assuming short-range atomic interactions and the Markovian approximation.59 It was also shown in 59 that thermostating only some of the degrees of freedom by using the Langevin dynamics brings the system, in equilibrium, to the corresponding canonical distribution. It can also be shown that the application of the Nose method to only a part of the system (i.e., to a subset of atomic degrees of freedom) also performs correct thermostating of the entire system to the target temperature.

The paper is organised as follows. In Sec. II we generalise the methodology of Refs. 42, 52 and 53 to the cases of two independent baths $\nu=1,2$ each having its temperature $T_\nu$. The generalisation, called GLE-2B, includes the use of two sets of auxiliary degrees-of-freedom (DOF) corresponding to each bath and their stochastic dynamics. This is performed by the use of a multi-variate Markovian stochastic process for the position and momentum of the DOF of the central region and the corresponding “position” and “momentum” of the two sets of auxiliary DOF59. The resulting algorithm is explicitly described in Appendix A and has been implemented in the code LAMMPS59 following our previous work on the GLE with a single bath.55 In Sec. III, we consider some applications of the GLE-2B for a specific realistic system. It consists of a short Al nanowire connected to two Al baths. Each bath is represented by a set of auxiliary DOF generated from a model solid, i.e., one half-sphere of an Al fcc lattice. In this first application of our GLE-2B approach, we concentrate on the steady-state properties of the system. First, we consider the equilibrium condition and an artificially thermally decoupled system to perform a first validation of our methodology. Then we treat the proper non-equilibrium conditions when $T_1 \neq T_2$. We also compare our results with other possible thermostating procedures. In Sec. IV, we interpret our results for the temperature profiles through the system in terms of the properties of integrable versus non-integrable cases. In the latter case, a full temperature gradient is established in the system, while in the former case there is no temperature gradient built up in the system. In terms of transport properties, a perfect thermal conductor (a ballistic thermal system) has an infinite conductance (in the thermodynamic limit) and, therefore, there is no temperature gradient within the central part of the system. Whenever the system presents some form of thermal resistance (finite conductance), a temperature gradient exists in the system. Finally, in Sec. V we conclude and discuss further developments of the present study.

FIG. 1. (Colour online) Schematic representation of the system. It includes the finite size central system (in blue) where the GLE dynamics is performed, and the two bath $\nu=1,2$ regions at temperature $T_\nu$. Because the forces $f_{\nu}$ and the quantities $g_{\alpha,\nu}$ are of finite range (not necessarily short ranged), one can perform the mapping of $\Pi_{b,b'}(\omega)$ on a finite region of space: the bath reduced region (in pink), one for each bath.

II. GENERALISATION TO THE TWO BATHS PROBLEM

A. Equations of motion for a system coupled to two baths: embedding Newton’s equations

We consider a central system (of finite size) interacting with two independent baths $\nu=1,2$ 55, see Fig. 1 for a schematic representation of the system. The corresponding classical Lagrangian is given by $\mathcal{L} \equiv \mathcal{L}_{\text{sys}} + \mathcal{L}_{\text{bath},(1)} + \mathcal{L}_{\text{bath},(2)} + \mathcal{L}_{\text{int}}$, where

$$\mathcal{L}_{\text{sys}}(\mathbf{r}, \dot{\mathbf{r}}) = \sum_{i=1}^{2N} \frac{1}{2} m_i \dot{r}_i^2 - V(\mathbf{r}) \quad (1)$$

$$\mathcal{L}_{\text{bath},(1)}(\mathbf{u}_1, \dot{\mathbf{u}}_1) = \sum_{i=1}^{L_1} \frac{1}{2} \mu_i \dot{u}_{i\gamma}^2 - V_{\text{harm}}(\mathbf{u}_1) \quad (2)$$

$$\mathcal{L}_{\text{int}}(\mathbf{r}, \mathbf{u}) = \mathcal{L}_{\text{int},(1)}(\mathbf{r}, \mathbf{u}_1) + \mathcal{L}_{\text{int},(2)}(\mathbf{r}, \mathbf{u}_2). \quad (3)$$

The positions of the atoms, labelled $i=1,2,\ldots,N$ with mass $m_i$, of the central system are given by vectors $\mathbf{r}$ with components $r_{i\alpha}$ ($\alpha$ indicating the appropriate Cartesian coordinate). $\mathcal{L}_{\text{sys}}$ is the Lagrangian of the system with potential energy $V(\mathbf{r})$. The Lagrangian $\mathcal{L}_{\text{bath},(1)}$ describes the harmonic bath (bath $\nu=1$). The bath’s atoms are labelled $l_1$ and have masses $\mu_{l_1}$. We introduce a shorthand notation for the labels of the bath degrees of freedom (DOF) $b_1 \equiv l_1 \gamma$, where $\gamma$ indicates the Cartesian coordinate. The corresponding potential energy $V_{\text{harm}}(\mathbf{u}_1)$ is harmonic with respect to the displacements $u_{l_1\gamma} = u_{b_1}$ of the bath atoms from their equilibrium positions. Its expression is

$$V_{\text{harm}}(\mathbf{u}_1) = \frac{1}{2} \sum_{b_1,b'_1} \sqrt{\mu_{l_1} \mu_{l_1}} u_{b_1} D_{b_1,b'_1}^{(1)} u_{b'_1}$$
where $D_{b_v,b'_v}^{(1)}$ are the elements of the dynamic matrix of the bath $\nu = 1$. The Lagrangian $L_{\text{bath},(2)}$ for the bath $\nu = 2$ is similar to $L_{\text{bath},(1)}$ and obtained from $L_{\text{bath},(1)}$ by swapping the bath index $1 \leftrightarrow 2$.

Finally the interaction between the central system and the baths is a linear superposition of the interaction between the system and each bath (we recall that the baths are independent). The individual contribution $L_{\text{int},(\nu)}(r,u_\nu)$ is taken to be linear in the bath displacements $u_\nu$ with the following expression:

$$
L_{\text{int},(\nu)}(r,u_\nu) = -\sum_{b_\nu} \mu_\nu f_{b_\nu}(r) u_{b_\nu} .
$$

Note that the dependence of such an interaction on the system DOF, via $f_{b_\nu}(r)$, remains arbitrary.

We can now derive the equations of motion for the central system DOF, that is in $\nu$. We consider the initial positions and velocities of the DOF of the bath

$$
m_i \ddot{r}_{i\alpha} = -\frac{\partial V}{\partial r_{i\alpha}} - \sum_{\nu=1}^2 \sum_{b_\nu} \mu_\nu g_{i\alpha,b_\nu}(r) u_{b_\nu} ,
$$

where $g_{i\alpha,b_\nu}(r) = \partial f_{b_\nu}(r)/\partial r_{i\alpha}$.

For the bath DOF, we find two sets of equations which can be solved analytically, since the Lagrangian $L_{\text{bath},(\nu)} + L_{\text{int},(\nu)}(r,u_\nu)$ is harmonic in the bath DOF $u_\nu$. These sets of equations are given by

$$
\mu_\nu \ddot{u}_{b_\nu} = -\sum_{b'_\nu} \sqrt{\mu_\nu \mu_{b'_\nu}} D_{b_\nu,b'_\nu}^{(\nu)} u_{b'_\nu} - \mu_\nu f_{b_\nu}(r) .
$$

for $\nu = 1,2$. Equation (6) can be solved by introducing the kernel of the differential equation defined from the eigenstates $v^{(\nu)}_{b_\nu}$ and eigenvalues $\omega_{b_\nu,\nu}^2$ of the dynamical matrix $D_{b_\nu,b'_\nu}^{(\nu)}$. The solution of Eq. (6) is then substituted into Eq. (5) to obtain a closed equation in terms of the system DOF only.

We consider the initial positions and velocities of the bath atoms, appearing in the solution of Eq. (6), being stochastic. It permits us to derive a generalised Langevin-like equation of motion (EOM) for the system DOF

$$
m_i \ddot{r}_{i\alpha} = -\frac{\partial V}{\partial r_{i\alpha}} - \int_{-\infty}^t \frac{d\tau}{\tau} \sum_{\nu,\nu',\alpha} K_{i\alpha,\nu',\alpha}(t,\tau;r) \dot{r}_{i\alpha}(\tau) + \sum_{\nu} \eta_{i\alpha}^{(\nu)}(t;r) .
$$

The dynamics of the system DOF is governed by an effective potential $\tilde{V}$, two memory Kernels $K_{i\alpha,\nu',\alpha}(t,\tau;r)$ and stochastic forces $\eta_{i\alpha}^{(\nu)}(t;r)$ corresponding to each independent bath $\nu$.

The potential energy $\tilde{V}$ is given by the nominal potential energy $V$ inside the central system plus the potential energy between the central system and the two frozen baths. There is also a “polaronic” correction energy due to the coupling between the system atoms and the harmonic displacements of the baths’ atoms around their equilibrium positions:

$$
\tilde{V}(r) = V(r) - \frac{1}{2} \sum_{\nu} \sum_{b_\nu,b'_\nu} \sqrt{\mu_\nu \mu_{b'_\nu}} f_{b_\nu}(r) \Pi_{b_\nu,b'_\nu}(0) f_{b'_\nu}(r) .
$$

The memory kernel for the bath $\nu$ is given by

$$
K_{i\alpha,\nu',\alpha}(t,t';r) = \sum_{b_\nu,b'_\nu} \sqrt{\mu_\nu \mu_{b'_\nu}} g_{i\alpha,b_\nu}(r(t)) \Pi_{b_\nu,b'_\nu}(t-t') g_{i\alpha,b'_\nu}(r(t')) .
$$

Finally, the stochastic (and hence non-conservative) forces $\eta_{i\alpha}^{(\nu)}(t;r)$ are functions of the initial positions and velocities of the DOF of the bath $\nu$. Following Refs. [42 and 52], we can now assume that each bath $\nu$, described by the combined Lagrangian $L_{\text{bath},(\nu)} + L_{\text{int},(\nu)}$, is in thermodynamic equilibrium at temperature $T_{\nu}$. Therefore, the stochastic forces $\eta_{i\alpha}^{(\nu)}(t;r)$ can be treated as random variables. From these assumptions, the dissipative forces are well described by a multi-dimensional Gaussian stochastic process with correlation functions.

We then introduce two sets of auxiliary DOF (aDOF) $\{s_{\nu,1}(k_\nu), s_{\nu,2}(k_\nu)\}$ corresponding to each independent bath $\nu$, taking the initial positions and velocities of the auxiliary DOF $\{s_{\nu,1}(k_\nu), s_{\nu,2}(k_\nu)\}$ to be zero.
bath $\nu = 1, 2$. They are associated with the corresponding mapping coefficients $\{\tau_{k_1}, \omega_{k_1}, \xi_{k_1}\}$ with $k_1 = 1, 2, \ldots, N_a^{\text{DOF}}$ and $\{\tau_{k_2}, \omega_{k_2}, \xi_{k_2}\}$ with $k_2 = 1, 2, \ldots, N_b^{\text{DOF}}$.

Note that the frequencies $\omega_{k_\nu}$ used in the mapping Eq. (13) of the matrix $\Pi_{b_\nu, b'_\nu}$ are not directly related to the eigenvalues $\omega_{\nu, \lambda}^2$ of the dynamical matrix $D_{b_\nu, b'_\nu}$, as explained in detail in Ref. [53]. There would be a one-to-one correspondence only when the number of aDOF, $N_a^{\text{DOF}}$, is exactly equal to the number of eigenvalues.

For a memory kernel of the type given in Eq. (13), solving the GLE Eq. (7) is equivalent to solving the following extended variable dynamics[22]

$$\dot{r}_{i\alpha} = p_{i\alpha}/m_i$$
$$\dot{p}_{i\alpha} = -\frac{\partial V}{\partial r_{i\alpha}} + \sum_{b_1, k_1} \frac{\mu_{1}}{\mu_{1}} g_{i\alpha,b_1}(r) c_{b_1}^{(k_1)} s_{1,1}^{(k_1)}(t) + \sum_{b_2, k_2} \frac{\mu_{2}}{\mu_{2}} g_{i\alpha,b_2}(r) c_{b_2}^{(k_2)} s_{2,1}^{(k_2)}(t)$$

$$s_{\nu,1}^{(k_\nu)} = -\frac{s_{\nu,1}^{(k_\nu)}}{\tau_{\nu}} + \omega_{\nu} s_{\nu,2}^{(k_\nu)} - \sum_{i, b_\nu} \sqrt{\mu_{b_\nu}} g_{i\alpha,b_\nu}(r) c_{b_\nu}^{(k_\nu)} \dot{r}_{i\alpha} + \sqrt{2k_B T_{\nu} \mu_{\nu}} \xi_{\nu,1}^{(k_\nu)}(t) \quad \text{for } \nu = 1, 2$$

$$s_{\nu,2}^{(k_\nu)} = -\frac{s_{\nu,2}^{(k_\nu)}}{\tau_{\nu}} - \omega_{\nu} s_{\nu,1}^{(k_\nu)} + \sqrt{2k_B T_{\nu} \mu_{\nu}} \xi_{\nu,2}^{(k_\nu)}(t) \quad \text{for } \nu = 1, 2$$

The corresponding total vector state $X = X[r_{i\alpha}, p_{i\alpha}, s_{1,1}^{(k_\nu)}, s_{1,2}^{(k_\nu)}, s_{2,1}^{(k_\nu)}, s_{2,2}^{(k_\nu)}]$ follows a multivariate Markovian process, where $\xi_{\nu,x}^{(k_\nu)}$ are independent Wiener stochastic processes with (white noise) correlation functions

$$\langle \xi_{\nu,x}^{(k_\nu)}(t) \rangle = 0$$
$$\langle \xi_{\nu,x}^{(k_\nu)}(t) \xi_{\nu',x'}^{(k'_{\nu'})}(t') \rangle = \delta_{\nu,\nu'} \delta_{xx'} \delta_{k_\nu, k'_{\nu'}} \delta(t - t') \quad (15)$$

We recall that even if the total vector state $X$ corresponds to Markovian processes, a subset of its components, for example the vector $X = X[r_{i\alpha}, p_{i\alpha}]$, is not necessarily following Markovian processes. This was clearly shown in the previous section where the random noise of the corresponding GLE is actually a coloured noise given by the memory kernel of the GLE. For such classes of non-Markovian processes (that are components, or functions of one or more components, of multivariate Markovian processes) self-consistency is guaranteed as the Chapman-Kolmogorov equation is satisfied[21].

C. Integration algorithm from a Fokker-Planck approach

Following the scheme given in Ref. [22], we now develop a Fokker-Planck (FP) approach to derive a set of equations which are equivalent to the equations of the extended Langevin dynamics given by Eq. (14).

We consider the probability distribution function (PDF) $P(r_{i\alpha}, p_{i\alpha}, s_{\nu,1}^{(k_\nu)}, s_{\nu,2}^{(k_\nu)}), t$ of the total vector state $X$. Such a PDF follows a FP dynamical equation which can be written as follows

$$\dot{P}(r_{i\alpha}, p_{i\alpha}, s_{\nu,1}^{(k_\nu)}, s_{\nu,2}^{(k_\nu)}), t) = -\dot{L}_{FP} P(r_{i\alpha}, p_{i\alpha}, s_{\nu,1}^{(k_\nu)}, s_{\nu,2}^{(k_\nu)}), t)$$

where $\dot{L}_{FP}$ is the corresponding FP Liouvillian.

We split the Liouvillian $\dot{L}_{FP}$ in two parts — a conservative part $\dot{L}_{\text{cons}}$ and a dissipative part $\dot{L}_{\text{diss}}$. The dynamics generated by the conservative part $\dot{L}_{\text{cons}}$ corresponds to the EOM of the DOF and aDOF given in Eq. (14) if one omits all the terms containing the $\tau_{k_\nu}$ parameters. The remaining dissipative part $\dot{L}_{\text{diss}}$ generates the EOM of the aDOF, given by the following generic form:

$$s_{\nu,x}^{(k_\nu)}(t) = -\frac{s_{\nu,x}^{(k_\nu)}(t)}{\tau_{\nu}} + \sqrt{2k_B T_{\nu} \mu_{\nu}} \xi_{\nu,x}^{(k_\nu)}(t) \quad (17)$$

For such a stochastic dynamics there exists an exact integration algorithm[22,23].

In order to obtain an integration algorithm (see details in Appendix A), we consider the time evolution of the PDF over an elementary time-step $\Delta t$

$$P(\ldots), t + \Delta t = e^{-\dot{L}_{FP} \Delta t} P(\ldots), t)$$

We use the splitting of $\dot{L}_{FP}$ and a second-order Trotter expansion to decompose the time evolution operator as follows[24]

$$e^{-\dot{L}_{FP} \Delta t} \sim e^{-\dot{L}_{\text{diss}} \Delta t_2} e^{-\dot{L}_{\text{cons}} \Delta t_1 e^{-\dot{L}_{\text{diss}} \Delta t_2}}$$

The first and last time evolution operators $e^{-\dot{L}_{\text{diss}} \Delta t_2}$ with half a time step $\Delta t/2$ generate steps (A) and (H) in the algorithm given in Appendix A.
Furthermore, we use a second Trotter expansion of the term $e^{-\mathcal{L}_{\text{cons}} \Delta t}$ by splitting $\mathcal{L}_{\text{cons}}$ in two parts $-\mathcal{L}_{r,s_1}$ and $\mathcal{L}_{p,s_2}$. The part in $\mathcal{L}_{p,s_2}$ generates the time evolution of the system DOF $p_{i\alpha}$ and of the aDOF $s_{\nu,2}^{(k_{\nu})}$ over half a timestep $\Delta t/2$, see steps (C) and step (G) in Appendix A. The part in $\mathcal{L}_{r,s_1}$ generates the time evolution of the system DOF $r_{i\alpha}$ and of the aDOF $s_{\nu,1}^{(k_{\nu})}$ over $\Delta t$, see steps (D) and (F) in Appendix A.

D. Calculation of the polarisation matrices $\Pi_{b_{\nu},b'_{\nu}}$

In order to perform the mapping given by Eq. (13), we first Fourier transform the equation into:

$$\Pi_{b_{\nu},b'_{\nu}}(\omega) = \sum_{k_{\nu}} \epsilon_{b_{\nu}}^{(k_{\nu})} \epsilon_{b'_{\nu}}^{(k_{\nu})} \left[ \frac{\tau_{k_{\nu}}}{1 + (\omega - \omega_{k_{\nu}})^2 \tau_{k_{\nu}}^2} \right. + \frac{\tau_{k_{\nu}}}{1 + (\omega + \omega_{k_{\nu}})^2 \tau_{k_{\nu}}^2} \right] \cdot (18)$$

For each bath, the set of parameters $\{\epsilon_{b_{\nu}}^{(k_{\nu})}, \tau_{k_{\nu}}, \omega_{k_{\nu}}\}$ is obtained from a fitting procedure (described in detail in Ref. [53]) based on the vibrational properties of the bath. As shown in Ref. [53], the polarisation matrices $\Pi_{b_{\nu},b'_{\nu}}(\omega)$ are related to the imaginary part of the phonon bath propagator $D_{b_{\nu},b'_{\nu}}(\omega)$ as follows:

$$\Pi_{b_{\nu},b'_{\nu}}(\omega) = -2\text{Im}D_{b_{\nu},b'_{\nu}}(\omega)/|\omega|.$$ 

The bath propagator is defined from the dynamical matrix $D^{(\nu)}$ of the bath $\nu$ as

$$D_{b_{\nu},b'_{\nu}}(\omega) = \left[ \omega^2 1 - D^{(\nu)} + i\varepsilon \right]^{-1}_{b_{\nu},b'_{\nu}}.$$ 

where $\varepsilon \rightarrow 0^+$. We use a real-space method, based on the Lanczos algorithm, to calculate the inverse of the matrix defining $D_{b_{\nu},b'_{\nu}}$ and the fitting procedure, described in Ref. [53], to get the values of the parameters $\{\epsilon_{b_{\nu}}^{(k_{\nu})}, \tau_{k_{\nu}}, \omega_{k_{\nu}}\}$.

Once the system is defined, the calculations of the dynamical matrices and the mapping procedures are performed individually for each bath $\nu$.

E. Generalisation to $N_{\text{bath}}$ independent baths

It should be noted that the generalisation to the case of the central system connected to $N_{\text{bath}}$ independent baths is straightforward. One can expand the GLE-2B by introducing $N_{\text{bath}}$ auxiliary sets of DOF $s_{\nu,1}^{(k_{\nu})}$ and $s_{\nu,2}^{(k_{\nu})}$. The EOM of these aDOF are given in the 3rd and 4th lines of Eq. (14). The EOM for the momenta of the central system will include the contribution of the $N_{\text{bath}}$ baths via the set of aDOF. It is defined by the sum over all the bath indices instead of just a sum over $\nu = 1,2$, i.e.

$$\dot{p}_{i\alpha} = -\partial V/\partial r_{i\alpha} + \sum_{\nu=1}^{N_{\text{bath}}} \sum_{b_{\nu},b'_{\nu}} \sqrt{\mu_{b_{\nu}}} g_{i\alpha,b_{\nu}}(x) \epsilon_{b_{\nu}}^{(k_{\nu})} s_{\nu,1}^{(k_{\nu})}(t).$$ 

FIG. 2. (Colour online) Model system for a short cylindrical Al wire (central system containing 59 atoms) connected to two left ($L$) and right ($R$) baths. For the GLE calculations, the $L$ and $R$ bath reduced region contains 68 atoms each. For the calculations of the dynamical matrix and the mapping of the polarisation matrix, larger baths (each containing 203 atoms) were considered. The embedded atom method (EAM) is used for the inter-atomic effective potential. For convenience, we now call the baths $\nu = L, R$ instead of $\nu = 1,2$.

The mapping and fitting procedures of the polarisation matrices $\Pi_{b_{\nu},b'_{\nu}}$ will be performed for each individual bath.

III. RESULTS FOR THE GLE-2B APPROACH

We now apply the GLE-2B approach to a model system, which consists of a short Al nanowire connected to two independent baths (represented by two half spheres) as shown in Figure 2. For obvious reasons, we now call the left and right baths $\nu = L, R$ (instead of $\nu = 1,2$). The electronic transport properties of similar Al nanowires have been studied some decades ago. It is now interesting to study their thermal transport properties under the proper non-equilibrium conditions.

Figure 3 provides more information about the central system (labelling of different layers of the system) and the bath reduced regions. The system is built using a fcc lattice and the distance between layer A and layer C (Fig. 3) corresponds to the lattice parameter of 4.05 Å. The Al nanowire (layers A to G) has the length of 12.15 Å. The central system (layers L3-L1, A-G, R1-R3) for which the GLE-2B simulations are performed has a length of 20.25 Å. We have taken the Embedded Atom Method (EAM) to model the metallic Al system. The tabulated interatomic potential is provided by the NIST Interatomic Potential Repository Project.
In order to compare to the results obtained with our GLE-2B approach, we also consider two different thermostatting approaches for the baths. These are more widely used and consist of stochastic dynamics for the atoms in the bath regions, while the central region follows the common deterministic Newtonian dynamics (see Fig. 3 and Appendix A). We consider two stochastic dynamics, i.e., a simple Langevin (Langevin-Gauss LG) dynamics and the dynamics generated by a Nose-Hoover (NH) thermostat which are already implemented in the MD code LAMMPS.

For the simple LG approach, the stochastic dynamics for the atoms in the bath regions are given by

$$\dot{p}_\nu = -\nabla_{r_\nu} V(r) - \gamma p_\nu + \xi^G_\nu$$

with the momentum vector $p_\nu$ of the atoms in the bath $\nu = L, R$ and the white noise vector $\xi^G_\nu$ (following a Gaussian distribution with a width related to the temperature $T_\nu$). The single parameter $\gamma$ characterises the friction ($\tau_\text{damp} = 1/\gamma$) for all atoms in the bath region.

For the NH thermostating approach, each is also characterized by a damping parameter $\tau_{\text{damp}}^{\text{NH}}$ (in unit of time) in the LAMMPS implementation.

As a first application of our GLE-2B, we concentrate on studying the steady-state properties of the system. For all stochastic dynamics, we consider some initial conditions (values of the baths’ temperatures $T_L$ and $T_R$ and of the velocities of the atoms in the central system) and let the system evolve in time until the total kinetic energy reaches a plateau, i.e., a constant value up to the corresponding thermal fluctuations. For the different GLE-2B calculations, the system takes roughly 80 ps to reach a steady state regime (i.e. $\sim 40000$ time steps for a $\Delta t = 2$ fs). For the LG and NH thermostat calculations, the steady state can be reached in less time-steps since the characteristic damping time is adjustable by the user.

**A. Systems at equilibrium**

In order to validate our GLE-2B approach and the corresponding algorithm, we first need to check that when the bath temperatures are equal, $T_L = T_R$, we obtain the correct results for the energy and/or the velocity distributions as expected from equilibrium statistical mechanics.

We have performed different calculations for different temperatures $T = T_L = T_R$ where $T = 100, 150, 200, 300, 400$ K. Figure 3 shows the velocity distribution functions for different atomic layers of the central system. The results for the velocity distributions show that the system indeed reaches the expected thermal equilibrium, where the velocity distributions follow the Maxwell equilibrium distribution.

It should be noted that for temperatures above $\sim 400$ K, our model system shows some structural instabilities. Therefore calculations are performed only with temperatures lower than 400 K.

![FIG. 3. (Colour online) Schematic representation of systems under consideration for the GLE calculations. The central system consists of 7 layers of Al (labelled A to G) and 3 extra layers embedded in the left bath (labelled L1, L2 and L3) and 3 extra layers embedded in the right bath (labelled R1, R2 and R3). The layer L3, L1, B, D, F, R1 and R3 (L2, A, C, E, G, R2) contains each 5 (4) Al atoms. The bath reduced regions are represented by the brown and blue regions for the left and right baths respectively. Each of these regions contains 68 atoms. We recall that, during the GLE simulations, the positions of the bath atoms are fixed. The left and right baths are at their own temperature $T_L$ and $T_R$ respectively. For the other thermostatting approaches, the bath reduced regions are described by their own stochastic dynamics (Langevin-Gauss (LG) or Nose-Hoover (NH) thermostatting) at their own temperatures $T_L$ and $T_R$. The central system (L1-L3, A-G, R1-R3) evolves according to a Newtonian (NVE) dynamics. The remaining outer atoms are kept at fixed positions to ensure the structural stability of the system in the LG and NH thermostatting calculations.](image1)

![FIG. 4. (Colour online) Histograms of the velocity distribution for different layers of atoms for the system shown in Fig. 3. (Top Left Panel) Layers L3, L2, L1 of the central system. (Top Right Panel) Layers R1, R2, R3 of the central system. (Bottom Panel) Layers A to G. The GLE calculations are performed for $T_L = T_R = 200$ K. All distributions (histograms) fit perfectly the equilibrium Maxwell distribution (solid red line) as expected.](image2)
To further complement the validity of our approach, we have also performed calculations for pseudo double equilibrium conditions. This is done by considering the two baths at different positions, and keeping fixed the atomic coordinates of the central layer D of the central system. This creates a thermal decoupling between the two sides, i.e. the frozen layer D acts as a perfect reflective barrier for the thermal transport between the two baths. Figure 5 shows the results obtained for the velocity distribution when the bath temperatures are $T_L = 200$ K and $T_R = 125$ K [76]. One clearly sees that one side of the system (layers L3, L2, L1, A, B, C) and layers (E, F, G, R1, R2, R3), fit perfectly the equilibrium Maxwell distributions obtained for the two temperatures $T_L$ (lines in the top left and bottom panels) and $T_R$ (lines in the top right and bottom panels).

To consider different temperatures $T_L \neq T_R$, it has been shown that a temperature gradient may or may not build up across the system. Temperature profile measurements are usually performed, by using thermal probe AFM, on mesoscopic scale systems (a few microns in length) or on multiwall carbon nanotubes (14 nm diameter and 4 microns length [79]). Such measurements have not yet been performed on nanoscale objects. However, a lot of theoretical work on model systems can be found in the literature. In the following paragraphs, we will review briefly the causes of the presence or absence of a temperature gradient as reported in previous theoretical studies.

Since the seminal work of Rieder et al. [77], it has been known that in an one-dimensional homogeneous harmonic system (also referred to as an integrable system), the thermal conductivity diverges in the thermodynamic limit. No temperature gradient is formed in the bulk of the system, since the dominating energy “carriers” are not scattered and propagate ballistically. A large variety of harmonic (integrable) systems have been studied in the classical [78,79] and quantum [80-81] limits, using analytical and/or numerical approaches. All these studies show that there is no temperature gradient inside the system (except small regions in the vicinity of the contacts between the central system and the baths). One usually obtains a constant-temperature profile, in the central system around the averaged temperature

$$T_{av} = (T_L + T_R)/2.$$  

On the other hand, in classical or quantum non-integrable systems, a temperature gradient is formed inside the system. The temperature gradient is uniform, and the heat conductivity is finite (it means that these systems obey Fourier’s law). The general trend is that one has to break the integrability of the system in order to build up a temperature gradient there. This can be done by introducing: (a) any form of anharmonicities, (b) extra local stochastic processes, (c) dephasing for quantum systems, or (d) mode coupling for classical systems.

The introduction of topological/configurational defects or disorder in harmonic systems can also lead to the build up of a temperature gradient. This result can be understood from the fact that defect/disorder introduces some form of localisation of the vibration modes [85]. Such modes do not favour ballistic transport, as phonons get scattered by impurities or boundaries. Hence the system has a finite thermal conductivity and presents a temperature gradient across itself.

We now test our GLE-2B method to investigate the thermal transport properties of the Al nanowire in the context of ballistic versus diffusive transport regimes. We consider different temperatures $T_L$ and $T_R$ for the left and right baths respectively. The temperatures are chosen value (up to the thermal fluctuations).

In a nanowire connecting two thermal baths when $T_L \neq T_R$, it has been shown that a temperature gradient may or may not build up across the system. Temperature profile measurements are usually performed, by using thermal probe AFM, on mesoscopic scale systems (a few microns in length) or on multiwall carbon nanotubes (14 nm diameter and 4 microns length [79]). Such measurements have not yet been performed on nanoscale objects. However, a lot of theoretical work on model systems can be found in the literature. In the following paragraphs, we will review briefly the causes of the presence or absence of a temperature gradient as reported in previous theoretical studies.

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First of all, we can see that the lineshape of the different velocity histograms for different atomic layers is shown in Appendix A. We then extract the corresponding local fitting the velocity histograms onto the Maxwell distribution. All distributions fit the local velocity histograms onto the Maxwell distribution for different layers of atoms for the system shown in Fig. 3. The GLE calculations are performed for proper non-equilibrium conditions with the bath temperatures being $T_L = 150$ K and $T_R = 50$ K. The lineshape of the different distributions (histograms) is similar to that of the equilibrium Maxwell-Boltzmann distribution. All distributions fit well onto the Maxwell distribution obtained from the average temperature $T_{av} = (T_L + T_R)/2 = 100$ K.

Figure 6 shows the velocity distribution for a non-equilibrium calculation performed with $T_L = 150$ K and $T_R = 50$ K. This is a typical result, and the following discussion can be applied to other combinations of bath temperatures $T_L$ and $T_R$ (not shown in the paper). The corresponding time evolution of the total kinetic energy of the central system and its statistical distribution is shown in Appendix A.

Second, all velocity histograms appear to follow the same equilibrium Maxwell distribution corresponding to the average temperature $T_{av} = (T_L + T_R)/2 = 100$ K (even for the layers L3 and R3 embedded in the L and R bath respectively). The overall results clearly indicate that, within such a non-equilibrium regime, we are dealing with an integrable system, essentially a complex harmonic system which perfectly transmits the heat flux between the two baths.

To simplify the analysis of our calculations, we now consider only the temperature profiles along the central system. Such profiles are calculated for each layer by fitting the local velocity histograms onto the Maxwell distribution. We then extract the corresponding local temperature for each layer.

Figure 7 shows the temperature profiles in the central system for different sets of temperatures $T_{L,R}$. From the procedure of fitting the velocity histograms to the Maxwell distribution, we estimate an uncertainty of approximately $\pm 5$ K on the temperature values. Even in the presence of such an error, one can see that there is no temperature gradient in the central system for the different temperatures considered in Fig. 7. This implies that, for such temperatures, the system behaves like a perfect harmonic (integrable) thermal conductor.

We can also compare our GLE-2B approach with the other LG and NH thermostatting approaches (Fig. 11). In such approaches, the atoms in the bath reduced regions are allowed to move and follow a dissipative dynamics ruled by a simple LG dynamics or by a NH thermostat. The atoms in the central region (while interacting with themselves and with the moving atoms of the two baths) follow a Newtonian NVE dynamics.

From the results shown in Figure 8, we can see that while the GLE-2B calculations provide a uniform temperature profile inside the system, the LG and NH thermostatting approaches display two different kind of behaviour depending on the chosen value of damping parameter. Note that in all LG and NH calculations, we have performed the calculations with as many time-steps as required to reach a stationary state. The larger the value of the damping parameter, the longer the dynamics is needed to reach the stationary state (See also Appendix A for the time evolution of the kinetic energy).

For small values of the damping parameter $\tau_{damp} = 7$ ps ($\tau_{damp}^{NH} = 0.7$ ps), we obtain a temperature gradient in the central system. The NH thermostats appear to provide an almost perfect linear gradient in the whole system, while the simple Langevin thermostatting corresponds to a small temperature gradient.

However for larger values of the damping parameter...
\( \tau_{\text{damp}} = 100 \text{ ps} \) (\( \tau_{\text{damp}}^{\text{NG}} = 20 \text{ ps} \)), both LG and NH thermostats provide a flat temperature profile. The LG calculations result in a temperature profile almost similar to that of the GLE-2B calculations.

It is interesting to see that changing the value of the damping parameter can lead to completely different physical results, i.e. the presence of a temperature gradient or its absence in the central system. Such a dilemma does not exist in our GLE-2B approach since it does not contain any adjustable parameter.

One should note that the small values of the damping parameter have been chosen in order to reproduce a relaxation of the kinetic energy (for the LG and NH thermostatting) similar to the relaxation obtained from the GLE-2B approach (See also Appendix [A]).

It seems fair to say that fitting the values of the damping parameter (to reproduce the evolution of the kinetic energy) is not enough for obtaining the same temperature profiles with the LG, NH thermostats and with the GLE-2B approach.

As mentioned above, the presence of a temperature gradient is a signature of the breaking down of the integrability of the system, and corresponds to a system with a finite thermal conductance. It also implies the introduction of anharmonic effects, configurational disorder or the introduction of other (uncontrolled) random processes.

In order to understand if one can obtain a temperature gradient with the GLE-2B approach, we have performed calculations in different situations where the harmonicity of the system is broken. This can be achieved by considering higher temperatures. In these cases, the atoms of the central system move in an effective potential well and consequently one can obtain a finite temperature gradient, as shown in Figure 8. One can see that larger deviations of the temperatures from the average \( T_{\text{av}} = (T_L + T_R)/2 \) are obtained on the side of the hotter bath. One can further increase such effects by introducing artificially some configurational disorder in the system. For that, we have considered the same system and at random we have given one atom in each layer L1, A-G and R1 a mass 20\% larger (smaller) than the mass of the other atoms in the system. The corresponding temperature profile is shown in Figure 9 and presents, as expected for such a disordered and/or anharmonic system, a finite temperature gradient. In Appendix [D] we present calculations for a model of a one-dimensional Al chain. The GLE-2B calculations obtained for such a simple model confirm, as expected, the analysis we have performed in this section.

Furthermore, it is also crucial to understand the importance of anharmonic effects. For that we introduce into the GLE-2B method some form of anharmonic effects in the baths by taking a finite life-time for the phonon modes of the bath. It means that vibrational excitations do not have an infinite life-time (as should be the case for an integrable harmonic system) but rather that they decay (dissipate) in time. The simplest way to treat such effects is to introduce a “self-energy” in the phonon bath propagator

\[
D_{b_v, b_v'}(\omega) = \left[ \omega^2 \mathbf{1} - \mathbf{D}^{(\omega)} + \omega \Sigma_{\text{anh}}^{(\omega)} \right]^{-1}_{b_v, b_v'}. 
\]
Furthermore, we consider that $\Sigma_{\text{anh}}$ is purely imaginary and simply modifies the linewidth of the spectral features in $\Pi_{b,b'}(\nu) = -2\text{Im}D_{b,b'}(\omega)/|\omega|$. In practice, such effects can be implemented in a rather straightforward way: once the mapping of $\Pi_{b,b'}(\omega)$ is established, we take the values of the fitted parameters $\tau_{k_b}$ and make them smaller. The features of the corresponding vibration modes in the phonon bath propagator are then broadened.

The results of such calculations, shown in Figure 10, suggest that anharmonic effects in the baths tend to favour the build up of the temperature gradient across the system.

Finally we would like to add that short nanowires can be ballistic (harmonic regime) for a range of temperatures $T_{L,R}$. However, such a behaviour might not be true for much larger (longer) and strongly heterogeneous systems. Indeed, in such systems, one may expect to observe the presence of disorder, of more localized vibrational modes or, more importantly, of vibrational mode “mixing” effects (interaction between phonons) which lead to the building up of a temperature gradient in the system. For example, the process of mode coupling has been studied in model three-dimensional systems and has been shown to lead to the presence of a temperature gradient in the system.

**IV. CONCLUSION**

We have developed a Generalised Langevin Equation (GLE) approach to treat non-equilibrium conditions when a central classical region is connected to two realistic thermal baths at two different temperatures. The method is called GLE-2B for Generalised Langevin Equation with two baths. Following the original GLE approach, the extended Langevin dynamics scheme is modified to take into account two sets of auxiliary degrees of freedom, each of which characterises the vibrational properties of the baths. These auxiliary degrees of freedom are then used to solve the non-Markovian dissipative dynamics of the central region. We have developed the corresponding algorithm for MD simulations and implemented it within the MD code LAMMPS.

As a first application, we have studied the heat transport properties of a short Al nanowire, that connects the left and the right Al baths, in the steady-state regime. We have mostly considered the establishment of a local temperature profile within the system when the two bath temperatures are different. Our results are interpreted in terms of the properties of harmonic versus non-harmonic systems, and the presence or the absence of defects. In agreement with earlier studies, we found that in a purely harmonic (ballistic) thermal conductor (with spatially extended normal modes), there is no temperature gradient across the central part of the system. Whenever the system presents some form of thermal resistance (finite conductance) due to anharmonic effects, disorder, or extra random processes, a temperature gradient is present in the system. Furthermore a concrete example of such effects in a model of a one-dimension Al chain is provided in Appendix D.

We have also compared the results of the simulations using the GLE-2B approach to the results of other simulations that were carried out using standard thermostatting approaches (based on Markovian Langevin and Nose-Hoover thermostats, see Fig. 11). In the latter cases, either a flat temperature profile or a temperature gradient across the central system can be obtained depending on the value used for the damping parameter. Upon the choice of this parameter, two different physical results can be obtained. Such a dilemma does not exist in the GLE-2B approach as it does not contain any adjustable parameters.

Furthermore, we have shown that the GLE-2B is able to treat, within the same scheme, two widely different transport regimes, i.e. systems which have ballistic (with no temperature gradient) or diffusive (with temperature gradient) thermal transport properties. This is a crucial point since the crossover between ballistic and diffusive transport regimes has been observed experimentally in organic molecules of different lengths connecting two electrodes, after having been predicted theoretically.

Penultimately we would like to add that the GLE-2B has also another advantage over the more commonly used thermostatting approaches. This method has been derived explicitly in order to be able to treat inherently non-equilibrium properties which cannot be simulated (in principle) by the NH thermostats. Furthermore, we have already shown in Appendix D of Ref. [52] that we can derive the GLE dynamics with a coloured noise which is not simply proportional to the memory kernel (as is the case for the classical limit of the equilibrium fluctuation-dissipation theorem). This means that quantum effects of the baths can be incorporated in the GLE dynam-
Appendix A: Verlet-type algorithm for the extended Langevin dynamics with two baths

Following the Markovian equations derived in Section II B and prescriptions given in Ref. [52], we use the following algorithm for a single time-step $\Delta t$.

The algorithm is derived, in a manner similar to the Verlet algorithm, from a different splitting and a Trotter-like decomposition of the total Liouvillian for the extended Langevin dynamics of the system DOF, $r_{i\alpha}$, and the auxiliary DOF $s_{i\alpha}^{(k_{\nu})}$ associated with the two independent baths $\nu = 1, 2$. Such a decomposition has been shown to provide a more appropriate description of the velocity correlation function.[50]

Algorithm:

(A) Randomise and propagate the aDOF
\[
s_{\nu,x}^{(k_{\nu})} \leftarrow a_{k_{\nu}} s_{\nu,x}^{(k_{\nu})} + b_{k_{\nu}} c_{\nu,x}^{(k_{\nu})}
\]
for all $k_{\nu}$ and $\nu = 1, 2$ and $x = 1, 2$.

(B) Calculate all $g_{i\alpha,b_{\nu}}(\{r_{i\alpha}\})$ from the derivatives of $f_{b_{\nu}}(\{r_{i\alpha}\})$

(C) Propagate the DOF and aDOF over $\Delta t/2$
\[
v_{i\alpha} \leftarrow v_{i\alpha} + \left(f_{i\alpha} + f_{i\alpha}^{\text{pol}} + f_{i\alpha}^{\text{GLE}}\right) \frac{\Delta t}{2m_{i}}
\]
\[
s_{i\alpha}^{(k_{\nu})} \leftarrow s_{i\alpha}^{(k_{\nu})} - \omega_{\nu} s_{i\alpha}^{(k_{\nu})} \frac{\Delta t}{2}
\]

(D) Propagate the DOF over $\Delta t$
\[
r_{i\alpha} \leftarrow r_{i\alpha} + v_{i\alpha} \Delta t
\]

(E) Recalculate all $g_{i\alpha,b_{\nu}}(\{r_{i\alpha}\})$ from the derivatives of $f_{b_{\nu}}(\{r_{i\alpha}\})$

(F) Propagate the aDOF over $\Delta t$
\[
s_{i\alpha}^{(k_{\nu})} \leftarrow s_{i\alpha}^{(k_{\nu})} + \left(\omega_{\nu} s_{i\alpha}^{(k_{\nu})} + f_{k_{\nu}}\right) \Delta t
\]

(G) Propagate the DOF and aDOF over $\Delta t/2$
\[
v_{i\alpha} \leftarrow v_{i\alpha} + \left(f_{i\alpha} + f_{i\alpha}^{\text{pol}} + f_{i\alpha}^{\text{GLE}}\right) \frac{\Delta t}{2m_{i}}
\]
\[
s_{i\alpha}^{(k_{\nu})} \leftarrow s_{i\alpha}^{(k_{\nu})} - \omega_{\nu} s_{i\alpha}^{(k_{\nu})} \frac{\Delta t}{2}
\]

(H) Randomise and propagate all the aDOF
\[
s_{i\alpha}^{(k_{\nu})} \leftarrow a_{k_{\nu}} s_{i\alpha}^{(k_{\nu})} + b_{k_{\nu}} c_{i\alpha}^{(k_{\nu})}
\]

where the different forces, $f_{i\alpha}, f_{i\alpha}^{\text{pol}}, f_{i\alpha}^{\text{GLE}}, f_{k_{\nu}}^{\text{GLE}}$ are explained below. The force
\[
f_{i\alpha} = -\frac{\partial V(r)}{\partial r_{i\alpha}}
\]
is the force acting on the system DOF $i\alpha$ due to the interaction between the atoms in the system and in the bath region(s); the “polaronic” force $f_{i\alpha}^{\text{pol}}$
\[
f_{i\alpha}^{\text{pol}} = \sum_{\nu} \sum_{b_{\nu},b_{\nu}'} \beta_{\nu}^{\nu} \Pi_{b_{\nu}b_{\nu}'} g_{i\alpha,b_{\nu}}(0) f_{b_{\nu}'}
\]
\[
= \sum_{\nu} \sum_{b_{\nu},b_{\nu}'} \beta_{\nu}^{\nu} \Pi_{b_{\nu}b_{\nu}'} g_{i\alpha,b_{\nu}} c_{b_{\nu}'}^{(b_{\nu})} f_{b_{\nu}'}
\]

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(with \( b_{\nu} \equiv l_{\nu} \gamma \) for DOF the \( \nu \)-th bath) is the force acting on the system DOF \( i\alpha \) due to the interaction between the system and bath regions which induces a displacement of the positions of the harmonic oscillators characterising the baths. In Eq. (A3), we used the fact that \( \Pi_{b\nu,b\nu}^n(0) \) is the inverse Fourier transform (evaluated at \( \tau = 0 \)) of \( \Pi_{b\nu,b\nu}^{\nu}(\omega) \) given by Eq. (15).

The force \( f_{i\alpha}^{\text{GLE}} \) acts on the system DOF \( i\alpha \) and arises from the generalised Langevin equations:

\[
f_{i\alpha}^{\text{GLE}} = \sum_{\nu} \sum_{b_{\nu},b_{\nu}'} \sqrt{\frac{\mu_{\frac{1}{2}}}{\mu_{\nu}}} g_{i\alpha,b_{\nu}}(\{r_{i\alpha}\}) c_{b_{\nu}}(k_{\nu}) s_1^{(k_{\nu})}. \tag{A4}
\]

The force \( f_{k_{\nu}}^{\text{GLE}} \) acts on the aDOF \( s_1^{(k_{\nu})} \) and also arises from the generalised Langevin equations

\[
f_{k_{\nu}}^{\text{GLE}} = -\sum_{i\alpha,b_{\nu}} \sqrt{\frac{\mu_{\frac{1}{2}}}{\mu_{\nu}}} g_{i\alpha,b_{\nu}}(\{r_{i\alpha}\}) c_{b_{\nu}}(k_{\nu}) v_{i\alpha} \tag{A5}
\]

The integration of the dissipative part of the dynamics of the aDOF (see steps (A) and (F) in the algorithm) includes the coefficients \( a_{k_{\nu}} = \exp(-\Delta t/2k_{\nu}) \) and \( b_{k_{\nu}} = [k_B T_{\nu} \mu_{\nu} (1 - a_{k_{\nu}})]^{1/2} \) and the uncorrelated random variable \( \xi_{1,2}^{(k_{\nu})} \) corresponding to the white noise.

### Appendix B: Evolution and statistics of the energy

In this Appendix we show the time evolution of the kinetic energy and potential energy for a system under nonequilibrium conditions. We also calculate the statistical distribution of the kinetic energy and briefly discuss the convergence of the numerical calculations for the GLE-2B and LG approaches.

Figure 12 shows the time evolution of the total kinetic energy \( E_{\text{kin}}^{\text{TOT}} \) and of the total potential energy \( E_{\text{pot}}^{\text{TOT}} \) of the system shown in Fig. 2 versus time (\( \Delta t = 2 \) fs) \[11\]. The calculations are performed with the bath temperatures \( T_L = 150 \) K and \( T_R = 50 \) K. The horizontal dashed blue lines correspond to the averages of the energy over the time range 100 to 300 ps. The convergence is fairly well obtained after \( \sim 80 \) ps (40000 timesteps). (b): Evolution of the total kinetic energy \( E_{\text{kin}}^{\text{TOT}} \) of the central system only (59 atoms from layers L3 to R3). For the LG calculations, the convergence towards the stationary value occurs after roughly 80 ps for the damping time \( \tau_{\text{damp}} = 7 \) ps. The stationary state is reached after more timesteps, as expected, for the larger damping time \( \tau_{\text{damp}} = 100 \) ps (see inset where it is apparent that the stationary value is reached after 500 ps or 250000 timesteps).

The influence of the value of \( \tau_{\text{damp}} \) is reflected in the distribution of the kinetic energy shown in Figure 13. Only the LG calculations performed with a large value of \( \tau_{\text{damp}} \) reproduce the distribution of \( E_{\text{kin}}^{\text{TOT}} \) obtained from the GLE-2B approach. The LG calculations performed with \( \tau_{\text{damp}} = 7 \) ps produce a much broader distribution.

Furthermore, in order to obtain the correct temperature profile given by the GLE-2B approach, one needs to use a large \( \tau_{\text{damp}} \) in the LG approach. The LG calculations made with \( \tau_{\text{damp}} = 7 \) ps, which result in a gradient in the temperature profile (i.e. a completely different physics behaviour, see Fig. 8), show a time evolution of \( E_{\text{kin}}^{\text{TOT}} \) that is much more similar to that obtained from the GLE-2B method. These results confirm that the choice of the adjustable parameter for the LG method
is crucial for being able to simulate the proper physical behaviour. This is also true for the NH thermostatting approach (results not shown).

Appendix C: Influence of the coupling to the baths and the system size

In this section, we consider another way of coupling the Al nanowire to the thermal baths. We treat a system similar to the system shown in Figure 2 and Figure 3 by including into the central system only the 7 layers labelled A to G (system containing 31 atoms). The atoms of the layers L1, L2 and L3 (R1, R2 and R3) have now been incorporated in the L (R) bath regions themselves.

For this system, we obtain the same physical results for the temperature profile as those presented in the main text. For low temperatures and small $\Delta T = T_L - T_R$, the central system is harmonic and no temperature gradient is obtained across the system, see Figure 14.

Once more, the LG thermostats provide two different temperature profiles depending upon the value of the damping parameter $\tau_{\text{damp}}$. For small values of the damping parameter ($\tau_{\text{damp}} = 7$ ps), we obtain a small temperature gradient in the central system while for larger values of the damping parameter $\tau_{\text{damp}} = 100$ ps, the LG thermostats provide a flat temperature profile.

In terms of convergence versus the baths size, one should first notice that, in the GLE-2B approach, we do not choose which group of atoms are connected to a thermal bath, as this is usually performed when using more conventional LG or NH thermostats.

The coupling of the central system to the thermal bath $\nu$ is obtained by the coupling of the DOF $l_\alpha$ to the aDOF $s^{(k_\nu)}_{\nu,1}$ via the matrix elements $g_{i\alpha,b_\nu}$, see Eq. (19) and Eq. (14). Such a coupling exists only when the matrix element $g_{i\alpha,b_\nu}$ is non-zero. We recall that $g_{i\alpha,b_\nu}(r) = \nabla_{f_{b_\nu}}(r)/\nabla_{l_\alpha}$ is the derivative, with respect to the coordinate of the DOF $l_\alpha$, of the force $f_{b_\nu}$ felt by the atom $l_\nu$ of the bath $\nu$ with DOF $b_\nu \equiv l_\nu\gamma$.

The range of the matrix elements $g_{i\alpha,b_\nu}$ is determined by the cut-off of the inter-atomic potential used in the calculations. The range of the quantities $g_{i\alpha,b_\nu}$ is actually smaller than the cut-off of the inter-atomic potential, as the former is the second derivative (versus spatial coordinates) of the latter.

Hence there is no need to increase the size of the reduced bath region, as long as DOFs of the central system are properly coupled to the existing atoms in the reduced bath region. Indeed, adding atoms in the reduced bath region (which corresponds to $g_{i\alpha,b_\nu} = 0$ elements) will not change the dissipative dynamics of the atoms of the central region.

However, one should not forget that the infinite spatial extension of the baths has already been taken into account. In particular, the continuous vibrational spectra of the infinite baths has been obtained through the calculation of their dynamical matrix and subsequently in the mapping procedure described in Sections II B and III D (for more detail, see also Ref. [53]).

The coupling to the thermal baths comes directly from the construction of the geometry of the system itself and from the cut-off of the inter-atomic potential. It is not controlled by the user as usually done with LG or NH thermostats. This is, once more, one of the main differences between the consistent (and more elaborate) GLE-2B approach and the main-stream LG or NH thermostatting approach.

As far as the convergence versus the size of the central system is concerned, we have already mentioned in the main text that nonballistic transport properties can be obtained for much longer and heterogeneous systems. The presence of long wavelength acoustic modes and their indirect coupling (via the baths) with other delocalised vibrational modes...
can also lead to the establishment of a more diffusive transport property associated with the presence of a temperature profile in the central system. The study of the transport properties versus the size of the central system is important, but out of the scope of the present paper, and will be treated elsewhere.

**Appendix D: A one-dimensional toy model**

In this section, we consider a toy model for a one-dimensional system: a chain made of 11 Al atoms connected to two baths as shown in Fig. 15. This is a simpler system than the three-dimensional wire considered in the main body of the paper. We use the GLE-2B approach for the dynamics of the central chain and study the local temperature profile of the perfect chain and of the chains containing one or two defects. To model the defect, we simply change the mass of the corresponding atom while conserving the same EAM inter-atomic potential for all atoms in the GLE-2B calculations.

To make the system simpler, we constrain the dynamics of the atoms in the central chain to a purely one-dimensional problem, i.e., the atoms can move only along the main axis (the $x$-axis) of the chain. Therefore each atom in the central region is associated with only one degree of freedom.

For a system at equilibrium, $T_L = T_R$, the GLE-2B calculations provide a velocity distribution for each atom of the central region which follows, as expected, the statistical Gaussian distribution of a single degree of freedom given by

$$f_{1D}(v_x) = \frac{m}{\sqrt{2\pi kT}} \exp \left( -\frac{mv_x^2}{2kT} \right). \tag{D1}$$

For the non-equilibrium conditions, $T_L \neq T_R$, the velocity distributions, calculated from the GLE-2B, for the atoms of the central chain also follow the lineshape of a Gaussian distribution. However the associated temperature varies across the chain.

Figure 16 shows the temperature profile across the chain when the masses of all atoms in the central chain are equal. For "low" temperatures $T_{L,R}$, the homogeneous system behaves as a harmonic system and one obtains a flat temperature profile as expected for integrable systems. For "higher" temperatures, the motion of some of the atoms sample the anharmonic part of the interatomic potential, and therefore the system is non-integrable and a temperature gradient is built up.

![FIG. 15. (Colour online) Model for a one-dimensional Al chain (central system containing 11 atoms labelled A to K) connected to the left ($L$) and right ($R$) baths. For the GLE calculations, the $L$ and $R$ bath reduced region contains 30 atoms each. For the calculations of the dynamical matrix and the mapping of the polarisation matrix, larger baths (containing each 95 atoms) were considered. The mapping of the polarisation matrix elements is done using 78 aDOF with a set of 78 parameters $\{c_{\nu(k)}, \tau_{\nu(k)}, \omega_{\nu(k)}\}$ for each bath $\nu = L, R$.](image)

![FIG. 16. Temperature profiles across the central 1D chain for different bath temperatures $T_{L,R}$. The labels Lbath (Rbath) represent the temperatures at the left (right) bath. The other labels A to K indicate different atoms in the central chain. For "low" temperatures $T_{L,R}$, the system is harmonic and one gets a flat temperature profile as expected for integrable systems. For "higher" temperatures, the motion of some of the atoms sample the anharmonic part of the interatomic potential, and therefore the system is non-integrable and a temperature gradient is built up.](image)
FIG. 17. (Colour online) Temperature profiles across the central 1D chain for different defects present in the chain. The labels Lbath (Rbath) represent the temperatures at the left (right) bath. The other labels A to K indicate different atoms in the central chain. The mass of atom H in Fig. 15 is increased by 10, 20, 30 or 40%. The case of two defects (atoms C and H with their mass increased by 40%) is also shown. For the temperatures considered, the chain is in the harmonic regime. However the presence of defects, and their associated more localised vibration modes, leads to the build up of a temperature gradient across the chain.

Thus, the calculations shown here for a toy model of a one-dimensional Al chain present the same qualitative physics as that obtained for the three-dimensional short wire described in the main text.

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40 I. Snook, The Langevin and Generalised Langevin Approach to the Dynamics of Atomic, Polymeric and Col-
We have used the data file Al99.eam.alloy from M. S. Daw and M. I. Baskes, Phys. Rev. B 59, 3393 (1999).

It should be noted that there cannot be any form of classicalization of the system when the system is uniformly thermostatted at $T_L = T_R$. From the results of these calculations, we observed that the Al nanowire melts down at $T = 500$ K for the EAM inter-atomic potential we used. During the MD run, we observe the formation of the smaller constriction in the middle of the nanowire and eventually all the atoms of the layers A to F end up on the surfaces of the L and R bath regions. At temperatures $T = 400 - 450$, we also obtain the formation of a smaller constriction in the middle of the nanowire (with atoms moving towards the left and right sides of the constriction). The constriction consists of 3 layers of only 2 atoms in each layer. This is what we call "structural instabilities", meaning that the central system does not have the same atomic configuration (layers made of 4 or 5 atoms). Such strong modifications of the configuration of the system make it difficult to perform a consistent analysis of the (layer-by-layer) temperature profiles.

One should notice that the melting of Al surfaces, depending on the Miller indices (hkl), may occur even at temperatures which are 150 to 200 K below the bulk melting temperature [see for example P. Stoltze, J. K. Nørskov, and U. Landman, Phys. Rev. Lett. 61, 440 (1988) ], i.e. roughly around 730 - 780 K. The coordination of an Al atom at the surface is roughly half of that of an atom in the bulk. Therefore we believe that it is not surprising that the melting of the nanowire happens at even lower temperature as the coordination of the atoms inside the nanowire is even smaller that the coordination of surface atoms.

The velocity distributions are calculated from the averages of several realisations of each velocity distribution obtained at a given timestep, after the system is thermalised. For $T = 200$ K, the thermalisation is reached after 78 ps of GLE dynamics, i.e. after 40000 time-steps with $\Delta t = 2$ fs. The distributions are averaged over all the time-steps in the time range $t = [78, 144]$ ps.

Calculation using the LG thermostats and the GLE-2B approach were performed at equilibrium when the system is uniformly thermostatted at $T_L = T_R$. From the results of these calculations, we observed that the Al nanowire melts down at $T = 500$ K for the EAM inter-atomic potential we used. During the MD run, we observe the formation of the smaller constriction in the middle of the nanowire and eventually all the atoms of the layers A to F end up on the surfaces of the L and R bath regions. At temperatures $T = 400 - 450$, we also obtain the formation of a smaller constriction in the middle of the nanowire (with atoms moving towards the left and right sides of the constriction). The constriction consists of 3 layers of only 2 atoms in each layer. This is what we call "structural instabilities", meaning that the central system does not have the same atomic configuration (layers made of 4 or 5 atoms). Such strong modifications of the configuration of the system make it difficult to perform a consistent analysis of the (layer-by-layer) temperature profiles.

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The atoms of the bath reduced regions are present in the GLE-2B method (implemented in the LAMMPS code), as one needs to calculate the matrix elements $g_{i\alpha,b\nu}(r)$ for the instantaneous atomic positions of the central system. However, the atoms in the bath reduced regions are kept fixed during the GLE-2B runs, hence they do not contribute to the total kinetic energy of the system shown in Fig. 2. Furthermore, the contribution to the total potential energy, from these atoms is just a constant throughout the GLE-2B runs.

The GLE-2B runs, as well as the LG runs with $\tau_{damp} = 7$ ps, are performed for 120 ps (120000 timesteps with $\Delta t = 1$ fs). The steady state is reached around 80 ps. The velocity histograms (from which the local temperature is extracted) are obtained from statistical averaging over the time range 100 to 120 ps. For the LG runs with $\tau_{damp} = 100$ ps, the steady state is reached after a longer time, $\sim 300$ ps. The velocity histograms are obtained from statistical averaging over the time range 450 to 600 ps.